SHORT STRUCTURAL PAPERS

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Cannizzarite

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Abstract. $Pb_{46}Bi_{54}S_{127}$ (Vulcano, Italy), monoclinic, $P2_1/m$, $a = 189 \cdot 8$ (2), $b = 4 \cdot 09$ (1), $c = 74 \cdot 06$ (7) Å, $\beta = 11 \cdot 93$ (8)°, Z = 2, $D_x = 6 \cdot 95$, $D_m = 6 \cdot 7$ Mg m⁻³. The structure is characterized by alternating layers A [(Pb,Bi)_{46}S_{46}] and B [(Pb,Bi)_{54}S_{81}] parallel to (001). Pseudotranslation according to statistical substructures in each layer is responsible for order-disorder phenomena.

Introduction. Graham, Thompson & Berry (1953) derived two monoclinic lattices for natural and synthetic cannizzarite from two systems of reflexions with relatively high intensity. These differed only by the relation $a_A: a_B = 7: 12$. The authors determined D_m with limited accuracy and found Bi, Pb and S to be the only chemical constituents. The chemical formula is based on these results combined with the present structure determination and the assumption of stoichiometry. Two different statistical substructures A $[(Pb,Bi)_{2}S_{2}]$ and B $[(Pb,Bi)_{4}S_{6}]$ were determined by Matzat (1972) (Fig. 1). Lattice relations were interpreted in connexion with structural data (Matzat, 1977). Cell dimensions of the sublattices are $a_4 = 4 \cdot 13$, $a_B = 7.03, b_A = b_B = 4.09, c_A = 15.48, c_B = 15.46$ Å, $\beta_A = 98.56, \beta_B = 98.00^{\circ}$. The common space group is P2/m.

Precise orthogonalization of data measured on a scanning microdensitometer from Weissenberg films (50 μ m raster) made it possible to order all observed reflexions on reciprocal-lattice points corresponding to the given unit cell, which can be described as 46 A and 27 B subcells combined in the a direction and penetrating each other as indicated in Fig. 1. Cell parameters and e.s.d.'s were determined within the orthogonalization process.

Space-group determination is based on the systematic absence of reflexions 0k0, k = 2n + 1 allowing $P2_1$ and the relative identity of intensities for *khl* and h(k + 2)l. When $I_{h0l} \approx I_{h2l}$, centres of Patterson peaks can be

located at y = 0 or $y = \frac{1}{2}$. Patterson density $p_{(xyz)}$ projected parallel to **b** will be $p_{0(xz)}$ calculated from I_{h0l} . The cosine-modulated projection $p_{2(xz)}$ calculated from I_{h2l} has identical relative values. Since $p_{0(xz)} = p_{2(xz)} \times \sum_{j} \cos (2\pi k y_j)$ summed over N peaks, the relation is valid for $y_j = 0, \frac{1}{2}$. Atom parameters of the structure are then y or $y + \frac{1}{2}$ corresponding to positions on mirror planes at $y = \frac{1}{4}$ and $y = \frac{3}{4}$, and allow the space group to be defined as $P2_1/m$. $I_{hkl} \approx I_{h(k+2)l}$ is valid in this case for all k. The difference between corresponding intensities is generally very small and given by a function of sin θ mainly determined by scattering factors and thermal parameters.

Weissenberg film data were collected with Cu K_{ll} radiation (graphite monochromator) from a crystal $0.036 \times 0.006 \times 0.5$ mm with exposure times of up to 20 d for layers h0l to h4l. The intensities of 2965 observable reflexions for layers h0l and h1l in the range (sin θ)/ $\lambda \le 0.625$ Å⁻¹ (13 954 accessible) were visually



Fig. 1. Cannizzarite. Sine-modulated electron density projection (h11) parallel to [010] of statistical substructures A and B in space group $P2_1/m$ and lattice parameters. The penetration of the combined substructures is indicated near a common 2_1 axis. Metal positions are named MA and MB, sulphur positions SA and SB, SO for substructures A and B respectively. Contours are at ± 16 , ± 32 and ± 64 e Å⁻².

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Fig. 2. (a) Projection of the structure of cannizzarite parallel to [010]. Substructural units of $(12A) \approx (7B)$ and $(17A) \approx (10B)$ are framed (compare b). One half of the (12A) plus the (17A) unit represents the contents of the asymmetric unit.



Fig. 2 (cont.). (b) Lattice geometry of cannizzarite. 2_1 axes common to substructures A and B are symbolized. Positional differences in the **a** direction for selected 2_1 axes of sublattices A and B are given in Å at positions where they nearly coincide (dashed symbol). Sublattice unit cells are shown around a common 2_1 axis as in Fig. 1.

estimated and corrected for Lorentz–polarization and absorption $[\mu(\operatorname{Cu} K\alpha) = 147.5 \text{ mm}^{-1}]$ with a program written by Paulus (unpublished), and were used for the structure determination.

A preliminary structure proposal was made by extending the combination of substructures (Fig. 1) in the **a** direction and defining coincident 2_1 axes as possible origins (Fig. 2b). Metal atoms of the *A* layer (*MA*) were displaced from the statistical position parallel to **c**^{*}, so that the sum of displacements added to zero and the smallest interatomic distance to the nearest S atom of layer *B* (SB) was the same for all metal atoms (3·17 Å). S atoms of layer *A* (SA) were displaced with the mean displacement of the neighbouring *MA*, coordinating them as a distorted square. Generally this kind of modulation was confirmed by the final results. Further refinement was achieved by Fourier syntheses. Full-matrix least-squares refinement of the metal positions revealed parameter shifts of up to 1 Å due to strong correlation between pseudotranslational parameters. A weighting factor of 0.05-0.1 was used to compensate for this effect and fairly good convergence was achieved. Fourier syntheses (atom positions and overall temperature parameter) revealed parameter shifts up to 0.15 Å. The final refinement was performed with difference syntheses (metal position and isotropic temperature parameters). Neutral-atom scattering factors with correction for anomalous dispersion for Bi, Pb and S (International Tables for X-ray Crystallography, 1974) yielded R = 0.099 for the observed reflexions. Reflexions F_{hkl} with indices $h \neq \pm n \times 46$ and $h \neq \pm n \times 27$ (n = 0, 1, 2 ...) can be interpreted as satellite reflexions due to atom position modulation in the substructures over a period of $46a_A$ or $27a_B$. The average intensity of such reflexions is smaller than that of sublattice reflexions, is dependent on the index h, and cannot be treated in the usual manner. Since the majority of the unobserved reflexions belong to this category it seems justified to scale them separately with a scale factor $\sum |F_{o(unobs)}| = \sum |F_{c(unobs)}|$. The residual R = 0.20 for all accessible structure amplitudes is then an indication that the modulation principle is correct. Final positional and thermal parameters are listed in Table 1. Interatomic distances and bond angles* and the occupation of metal positions by Pb and Bi based on these data are in good agreement with those in comparable Pb-Bi sulphosalt structures (Edenharter, 1976). Isomorphous replacement (Pb,Bi) has to be accepted at least as indicated in Fig. 2(a).

^{*} Lists of structure factors and interatomic distances and bond angles have been deposited with the British Library Lending Division as Supplementary Publication No. SUP 33797 (79 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

Table 1. Fractional atomic coordinates and isotropic thermal parameters (Å²) for cannizzarite

The estimated temperature factor for S atoms is B = 1.6 Å². E.s.d.'s are 0.006 Å and 0.1 Å² for metal atoms, M, and 0.03 Å and 0.3 Å² for S atoms. Atoms in layer A are named MA, SA, in layer B MB, SB and SO.

ATOM	x	Y	z	БA	TOM	x	۲	Z	8	ATOM		x	Y	z		B A	rom	x	۲	z	в
MA 1(PB)	.58432	1/4 .	2 21 3 4 6	.90 M	A 2 6 (PE	.42771	1/4	.39291	2.50	MB 5(P	в) .	04197	1/4	.1248	3 1.	80 M 8	30(81)	.506	50 3/4	.11876	2.40
MA 2(PB)	.91071	1/4 .	37001 2	•50 A	A27(B1	.43510	1/4	• 4 2 8 9 2	1.75	MB 6(B	I) .	05752	3/4	.1327	9 1.	85 ME	3 1 (PB)	.522	76 1/4	.12526	2.30
MA 3(BI)	.91783	1/4 .4	41511 2	•00 M	A28 (PE	.47666	1/4	.37628	2.45	MB 7 (P	·B) •(08013	1/4	•1231	0 1.	80 ME	332(B1)	.537	50 3/4	.13526	2.50
MA 4(PB)	.95152	1/4 •	56554 2	•35 M	A29(PE	.49487	1/4	.38679	2.35	MB 8 (B	D •	10208	3/4	.1129	1 1.	70 ME	33(PB)	.561	30 1/4	12269	1.90
MA 5(BI)	. 42640	1/4 .4	42201 1	•05 M	A 31 (PE		1/4	• 39611	3.29	MB 9(B	1) •	12459	1/4	.1016	7 2.	10 ME	334(BI)	• 582	35 3/4	•11544	1.45
MA 7(PB)	.02173	1/4 • .	37460 2	50 1	A 3 7 (61	5/70/	1/4	+ 16 77	2.93	MBIUIB		15594	3/4	1 7 5 7		90 MI	333(81)	.005	0 1/4	110500	2.20
MA S(BI)	02842	1/4	41.050 1	-70 K	A33(PE) .58767	1/4	.38442	2.60	MB12(B		16696	3/4	. 1363	1 1.	60 ME	37(81)	. 639/	29 3/4	.11000	2.50
MA 9(PB)	.05856	1/4	39129 2	40 M	A34(E)	3 .59048	1/4	.41982	2.30	MB13(P	· · · ·	19111	1/4	. 1220	19 1.	70 M	337(01)	.6512	87 3/4	.12722	2.15
MA10(BI)	. 36541	1/4 .4	42389 1	.35 M	A 35 (PE	.02697	1/4	.37858	2.45	MB14 (B	in li	21259	3/4	.1134	9 1.	80 ME	39(PB)	.671	32 1/4	.12570	2.30
MA11(PB)	.10718	1/4 .	37522 2	.70 M	A36 (P	.65240	1/4	.37065	3.00	MB15(B	ю.	23419	1/4	.1056	2 1.	70 ME	340(BI)	.687	2 3/4	.13104	1.85
MA12(P8)	.12766	1/4 .	38646 1	.90 M	A 37 (B 1) .058č5	1/4	.41006	2.06	мв16(ь	D .	24958	3/4	.1132	1 2.	00 ME	341(PB)	.7110	28 1/4	.11971	2.00
MA13(PE)	.13958	1/4 .4	46453 2	.70 M	A3c (PE	•68958	1/4	.38958	2.80	МВ17(Б	D .	26684	1/4	•1165	7 2.	10 ме	342(BI)	.7320	06 3/4	.11212	1.35
MA14(P5)	.16648	1/4 .	38707 2	•75 M	A 39 (BI) .09633	1/4	•42758	1.75	ма18(в	D •1	27966	3/4	•1317	6 2.	00 M E	343(BI)	.7544	1 1/4	.10139	1.85
MAIS(BI)	.17593	1/4 .4	42345 2	-00 M	A4C(PE	.73740	1/4	• 37516	2.60	MB19(P	в) .	30080	1/4	•1260	15 1.	90 ME	344(BI)	.765	71 3/4	.11951	2.10
MATC(PB)	.21378	1/4 .	50220 2	-45 M	A41(PE	.75870	1/4	.37873	1.90	MB20(B	D •	31948	3/4	•1257	8 1.	65 ME	345(PB)	.7824	2 1/4	.12486	2.40
MA12(81)	.22446	1/4 .4	41095 3	•50 M	A42(B]	.76848	1/4	-40860	1.90	M621(P	е) .	34186	1/4	.1167	6 1.	90 ME	346(BI)	.797	21 3/4	.13475	1.55
MAIC(PB)	-23013	1/4 • 2	2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2 2	.41) 17	A43(P6		1/4	• 39133	2.00	MB22(B	I) •	56504	5/4	•1085	2.	15 ME	347(PB)	• 821	19 1/4	.12130	2.00
MA70(PB)	20140	1/4 .	1111 2	•00 M	A44(B1	J .00522	1/4	.42/09	1.55	MB25(B		38394	1/4	.1020	4 2.	US ME	348(81)	•842	5 3/4	.11254	1.80
MAC1(PB)	.32042	1/4	3 12 2	.15 m	445(PE		1/4	.30207	3 45	MB24(B	• •	1 2 2 5	1/4	1267	7 2	00 ME 05 ME	50(01)	• 0044	27 1/4	11707	2.10
MA22(91)	.32751	1/4 .4	42523 1	.75 M	4 1 (BI) 7337	1/4	.11110	2.30	MB26(6	1)	2734	3/4	1241	0 1	65 MG	51(00)	80/	12 1/6	12266	2.20
*A23(PB)	.36760	1/4 .	37513 2	.50 M	a 2(B)) .99274	3/4	.10855	1.90	Ma27(P	A)	5087	1/4	.1214	6 1.	50 ME	52(BT)	. 9081	12 3/4	.13483	1.50
MA24(PB)	.39045	1/4 .	37426 2	.15 M	a 3(8)) .01253	1/4	.10441	2.30	MB28(6	л .	7229	3/4	.1128	1 1.	40 MF	353(PB)	.930	52 1/4	.12542	1.80
XA22(BI)	.39815	1/4 .4	41.37 1	.85 M	a 4(EI) .U2402	314	.12254	1.95	MB29(B	1) .	9394	1/4	.1041	0 2.	20 ME	354(BI)	.9500	59 3/4	.12122	2.25
ATGM	X Y	z	ATO	ч x	Y	z	ATOM	x	۲	z	ATOM	х	¢	¥	z	ATOM	x	۲	Z		
SA 1.35	583 3/4	.4358	33 5A2	7.419	17 3/4	.44292	SB 7	.04917	3/4	.22167	S 8 3 3	.530	083	3/4 .2	2000	SO 5	.08125	3/4	.00625		
SA 2 .89	083 3/4	.4004	1 SA2	.442	,2 3/4	.43542	SB 8	.77375	1/4	.20333	SB34	.552	92	1/4 .2	0917	SO 6	.09792	1/4	.01083		
SA 3 .90	0292 3/4	.4275	57 SA2	9.475	83 3/4	.49750	SB 9	.08917	3/4	.20875	S 8 3 5	.570	14 2	3/4 .2	1125	SO 7	.11958	3/4	.00333		
SA 4 .92	2750 3/4	.4179	2 SA3	0.427	50 3/4	• 4 3 3 3 3	S 8 1 0	.10458	1/4	.21666	S 6 3 6	•590	142	1/4 .2	0709	SO 8	•14042	1/4	• 99583		
SA 5 .94	4292 3/4	•4366	57 SA3	1.520	60 3/4	.49417	SB11	.12042	3/4	.22583	SB37	.605	83	3/4 .2	1458	SO 9	.16167	3/4	.98875		
SA 6 .96	5125 3/4	• 4 4 3 3	33 SA3	2 .532	08 3/4	•43125	SB12	.13792	1/4	.22917	SB38	•622	08	1/4 .2	2083	s010	.17708	1/4	.99625		
SA 7 .00	083 3/4	• 3975	18 SA3	3 .555	00 3/4	.42542	SB13	•16042	2 3/4	.22042	5839	•640	000	3/4 .2	2500	S011	•19292	3/4	.00417		
SA 8 .01	1292 3/4	.4254	2 SAS	4 • > / >	00 3/4	.43208	5814	+16292	1/4	-20792	5840	.00/	50	1/4 •2	2708	5012	.20750	1/4	.01417		
SA10 .02	2192 214 2958 314	. 4 4	27 SA3	5 • 5 9 C	UC 3/4	.39667	5816	-2175	1/4	.21250	5642	.703	75	1/4 .2	1910	5015	-22920	3/4	.00200		
SA11 .03	7167 3/4	. 4395	0 5A3	7 .643	75 3/4	.42375	SB17	.23375	3/4	.21875	SE43	.719	17	3/6 .2	0792	\$015	. 27250	3/4	.08017		
SA12 .1(3667 3/4	+405	53 SA3	8 .669	17 3/4	41167	5618	.24958	1/4	.22666	SB44	.735	00	1/4 .2	1500	5016	.28875	1/4	.99560		
SA13 .1	1833 3/4	.4320)5 5A3	9.680	دع 314	• 4 4 0 4 1	SB19	.27042	2 3/4	.22333	S845	.750	000	3/4 .2	2667	S017	.30583	3/4	.99875		
SA14 .14	4917 3/4	.4.175	92 SA4	0.701	25 3/4	.44167	5 B 2 O	.28917	1/4	.22167	S846	•767	'92 ·	1/4 .2	2875	SO18	.32167	1/4	.00667		
SA15 .10	5042 3/4	• 4 3 7 5	50 SA4	1.737	50 3/4	.40417	SB 2 1	.31125	5 3/4	.21375	5847	.789	58	3/4 .2	2208	S019	.34000	3/4	.00708		
SA16 .18	5333 3/4	• 4316	56 SA4	د 75 c	00 3/4	.43000	5622	.33292	2 1/4	.20417	5648	•812	92	1/4 .2	0667	S 0 2 0	.35667	1/4	•01125		
SA17 .20	5958 3/4	•4225	SU SA4	3 .778	75 3/4	•41042	SB23	•34875	3/4	.20833	5649	•829	17	3/4 .2	1042	S 0 2 1	.38042	3/4	.99959		
SA18 -2	2417 374	•438/	75 SA4	4 .788	55 574	.44375	5824	.30500	1 1/4	•21417	5650	.840	007	1/4 .2	1333	5022	•40125	1/4	.99208		
5A19 +20	5085 5/4 7/17 3/4	. 398/	() SA4	5 •813	33 3/4 Ns 3/4	.43291	5825	. 30709	2 1/4	.22439	5851	-800		1/4 .2	2203	5025	• 4 2 2 9 2	3/4	•98417		
5A2U +2	1411 3/4 0017 3/4	. 423	גיאנ אונ קרי קר	0 .342	25 2/4		5620	. 41049	2 374	.22125	5657	.000	142 :	3/4 .2	2208	5024	.43/08	7/4	.99342		
5422 . 3	1208 374	. 4123	22 58	2 .961	67 114	-20625	5828	-44333	5 1/4	20500	5854	.921	67	1/4 .7	1375	5025	.46833	1/4	-01043		
5423 -3	3062 3/4	. 4 4 4	17 SA	3 .977	50 3/4	.21083	5829	.45958	3 3/4	.20833	50 1	.011	25	3/4 .0	9542	5027	.49000	3/4	-00202		
SA24 .3	6917 3/4	400	33 SH	4 .993	75 1/4	.21708	SB 30	.47458	3 1/4	.21750	SO 2	.031	25	1/4 .9	9125			2/4			
SA25 .3	8167 3/4	.4270	D9 SB	5 .010	00 3/4	.22542	5831	.49083	5 3/4	.22542	SO 3	.049	58	3/4 .9	9125						
SA26 .4	0875 3/4	.411	25 SB	6 .027	08 1/4	.22875	SB32	.50792	2 1/4	.22958	SO 4	.066	25	1/4 .9	9667						

Discussion. In connexion with the structural investigation and synthesis of Pb-Bi sulphosalts the determination of the crystal structure of cannizzarite was undertaken. The structure shows a predominant pseudotranslational component of $\frac{17}{46} \approx \frac{10}{27}a$, which fits within 0.15 Å, when applied to the statistical subcells with $a_A = \frac{1}{46}a$ and $a_B = \frac{1}{27}a$. The atomic positions (mean difference 0.07 Å) and interatomic coordination are also comparable according to this translation, so it has to be introduced as a partial symmetry operation in addition to those given by Dornberger-Schiff (1966). The consequences will be discussed in a separate paper. With multiple application of this partial operation it is possible to derive substructural units of $\frac{17}{46}$, $\frac{12}{46}$, $\frac{5}{46}$... (e.g. $3 \times \frac{17}{46} - 1 = \frac{5}{46}$) or geometrically described in subcells of $(17A) \approx (10B)$, $(12A) \approx (7B)$.

These can be combined to form similar structures, all named cannizzarite but differing in unit-cell dimensions.

The determined structure is then comprised of geometrical units: $2 \times (17A) + (12A) \approx 2 \times (10B) + (7B)$. A synthetic specimen of cannizzarite kindly provided by Dr Mariolacos (Göttingen) shows lattice and intensity features which confirm the relative identity of the structures, but is probably combined of 41 A subcells and 24 B subcells or described in larger units of $2 \times (12A) + (17A) \approx 2 \times (7B) + (10B)$, with $a \approx 169$ Å. In Fig. 2 one possibility for separating the most probable larger structural and geometrical units of $(17A) \approx (10B)$ and $(12A) \approx (7B)$ is indicated. In accordance with the metal position occupation and pseudo 2_1 axes, space group $P2_1/m$ is probable for all cannizzarite structures.

The type of connexion between structural layers formally derived from the galena type is novel. Boundary planes of layers parallel to (001)(cannizzarite) are (100) and (111)(galena) of types A and B respectively;

the orientation parallel to [010](cannizzarite) is [110](galena) of both layers. The structure of phase 3 synthesized by Graham *et al.* (1953) can be proposed from the cannizzarite structure by adding one octahedral layer to the compact close-packed layer *B* in cannizzarite and this is in good agreement with the statistical lattice parameters given by the authors.

The author is grateful to Professor R. Huber and Dr P. Schwager for permission to use the scanning microdensitometer and for assistance during the measurements. All calculations were performed on a Univac 1108 computer of GWD, Göttingen, with programs written by the author, except the intensity reduction program.

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Ammonium Hexachlorotechnetate(IV)

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Abstract. $(NH_4)_2 TcCl_6$, Fm3m, cubic, a = 9.9072 (8) Å, U = 972.42 Å³, Z = 4, $D_x = 2.34$ Mg m⁻³, $\mu = 2.97$ mm⁻¹ (Mo $K\alpha$). The dianionic complex of technetium is octahedral with six Cl ligands bonded at a distance of 2.3531 (5) Å. The ammonium ions which occupy the eight tetrahedral sites ($\bar{4}3m$) surrounding each complex anion are rotationally disordered. The structure was refined to an R of 0.033 using 115 independent, observed reflections.

Introduction. As part of an ongoing study of Tc complexes as radiopharmaceuticals, we are precisely determining the structures of a series of Tc complexes.

A nearly octahedral, yellow crystal, with edges ~ 0.25 mm was mounted on a glass fiber. Intensities were measured using Mo K_{α} radiation on a Syntex PI diffractometer equipped with a graphite monochromator. The data supported the choice of space group Fm3m in agreement with the previous work of Schwocau (1964) and Elder, Fergusson, Gaines, Hickford & Penfold (1967) for various salts of hexahalotechnetium dianions. A total of 552 reflections ($2.5 < 2\theta < 63.8^{\circ}$) were measured. After corrections for absorption, 115 unique reflections were obtained by averaging equivalent reflections. All 115 had $I > 2\sigma(I)$ and were used in the subsequent refinement. Calculated transmission coefficients varied from 0.59 to 0.70.

The choice of space group requires the Tc atoms to be located at the fourfold position (0,0,0), the N atoms at the eightfold position $(\frac{1}{4}, \frac{1}{2}, \frac{1}{4})$ and the Cl atoms at the

24-fold position of the type (x,0,0). A sharpened, originremoved Patterson map was used to locate the Cl atoms (0.24,0,0) and confirm the choice of the N-atom positions. Refinement of a scale factor, the Cl positional and anisotropic thermal parameters and Tc and N isotropic thermal parameters (six parameters total) converged with R = 0.033 and $R' = [\sum w(\Delta F)^2/\sum w(F_0)^2] = 0.025$.

The H atoms could not be located by examination of electron density or difference maps. Various models were tried placing H at the 32-fold position (x,x,x) with x = 0.19 or x = 0.31. Since none of these resulted in improved agreement with the observations, we report here the results for the model without H atoms.* The refined parameters are as follows: Cl, x = 0.23751 (5), $U_{11} = 0.0173$ (3), $U_{22} = U_{33} = 0.0354$ (3) Å²; Tc, U = 0.0172 (2) Å²; N, U = 0.0315 (8) Å². Neutral-atom scattering factors from Cromer & Mann (1968) were corrected for anomalous dispersion (International Tables for X-ray Crystallography, 1974). A final difference synthesis was nearly featureless, exhibiting maxima of ~ 0.4 e Å⁻³ at 0.08, 0.08, 0 and 0.48, 0.12. 0. Calculated positions for H had difference densities of ~0.2–0.1 e Å⁻³. Minima were observed ~0.1 Å from Cl ($-1.4 \text{ e} \text{ Å}^{-3}$), $\sim 0.6 \text{ Å}$ from Tc ($-1.0 \text{ e} \text{ Å}^{-3}$) and at

^{*} A list of structure factors has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 33898 (2 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.